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## The Crystal Structure of Bis(diethyldithiophosphato)nickel(II), Ni [(C<sub>2</sub>H<sub>5</sub>O)<sub>2</sub>PS<sub>2</sub>]<sub>2</sub>

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Bis(diethyldithiophosphato)nickel(II) is monoclinic, space group  $P2_1/c$ , with two molecules per unit cell. Unit-cell dimensions, determined with the use of standardized silicon powder and computed by the method of least-squares, are  $a = 10.493 \pm 0.005$ ,  $b = 10.300 \pm 0.003$ ,  $c = 8.76 \pm 0.01$  Å,  $\beta = 102.59 \pm 0.05^\circ$ .

The structure has been determined from a three-dimensional Patterson synthesis and a series of three-dimensional Fourier syntheses. Refinement was by the method of full-matrix least-squares, using data of the type  $h0l-h7l$ , to the final residual of 0.115. The molecule itself is centrosymmetric with nickel at the centre of symmetry. The nickel and four sulphur atoms are in square planar configuration, with nickel-sulphur bonds of 2.23 and 2.24 Å, which are well within the range 2.1–2.3 Å found for other square planar nickel(II) complexes. The nickel atom shows no tendency to coordinate to any atom in other molecules.

### Introduction

This diamagnetic compound was first prepared by Malatesta & Pizzotti (1945) and since then consideration has been given to its probable molecular structural formula together with the structural formulae of similar complexes. Quite recently Lopez-Castro & Truter (1963) observed that in diamagnetic nickel(II) complexes the Ni-S distance is 2.1–2.3 Å, whereas in six-coordinate complexes it is 2.4–2.6 Å. They have commented that this is in accordance with ligand-field theory which predicts that the stronger field produced by the ligands closer to the nickel will cause the  $d_y$  electrons to pair in the  $d_{z^2}$  orbitals, so producing a diamagnetic complex with no close neighbours in the  $z$  direction. Hence the most likely structural configuration for this compound would be one shown in Fig. 1(a) with a nickel atom and four sulphur atoms in square planar configuration. Other configurations, shown in Fig. 1(b), (c) and (d), were considered possible but less likely.

This compound has, however, a few rather unusual properties:

(i) Four-membered rings like Ni-S-P-S in Fig. 1(a) are comparatively rare. When this investigation was commenced, only one such structure was fully determined (Barclay & Hoskins, 1962) but since then another similar structure (Franzini, 1963) has been reported.

(ii) Its deep purple colour is unusual, as most diamagnetic square-planar nickel compounds are yellow, brown or red.

(iii) It has a high solubility in most solvents including non-polar light petroleum.

These properties have raised doubts about the correctness of the theoretical predictions, and an investigation of its structure by X-ray methods was undertaken in order to resolve any uncertainties.

Since the completion of the structure determination we have learned of a preliminary communication regarding Ni[(OEt)<sub>2</sub>PS<sub>2</sub>]<sub>2</sub> by Glinskaya & Porai-Koshits (1959) which gave a space group, a unit cell, and the Ni-S distance and configuration in good agreement

with the present work. Their report appears, however, to have been based solely on a Patterson synthesis and a single electron-density projection. This is scarcely conclusive, particularly in view of the difficulty in distinguishing in projection between sulphur, phosphorus and two superimposed oxygen atoms.

### Experimental

#### Crystal data

Bis(diethyldithiophosphato)nickel(II)

$\text{Ni}[(\text{C}_2\text{H}_5\text{O})_2\text{PS}_2]_2$

Colour: deep purple, almost opaque.

Solubility: highly soluble in most organic solvents both polar and non-polar;  
soluble also in water and very slightly in oils.

Unit-cell dimensions at 20°C:

$$\begin{aligned} a &= 10.493 \pm 0.005 \text{ \AA} \\ b &= 10.300 \pm 0.003 \\ c &= 8.76 \pm 0.01 \\ \beta &= 102.59 \pm 0.05^\circ \end{aligned}$$

The errors quoted are four times the standard deviations.

Space group extinctions from Weissenberg photographs are

$$\begin{aligned} h0l: l &= 2n + 1 \\ 0k0: k &= 2n + 1, \end{aligned}$$

indicating the monoclinic space group  $P2_1/c$ .

Because of the high solubility of the substance, its density was estimated by measuring the dimensions and mass of a well-developed crystal. This gave a value of  $1.4 \text{ g.cm}^{-3}$ . The calculated density for  $Z=2$  is  $1.54 \text{ g.cm}^{-3}$ . With  $Z=2$ , the nickel must be in a special position at the centre of symmetry; the molecule itself is centrosymmetric with one-half of each molecule occupying one of the four general positions of the space group.

For Cu  $K\alpha$  radiation the linear absorption coefficient  $\mu = 71.4 \text{ cm}^{-1}$ .

Accurate unit-cell dimensions were obtained from  $h0l$  and  $hk0$  Weissenberg photographs obtained with crystals sprinkled with silicon powder\* as a calibrating substance. The cell parameters and their standard deviations were obtained by a least-squares program written by Sanger (1964), and 45 reflexions with resolved  $\alpha_1$  and  $\alpha_2$  spots lying between the 533 and 444 powder lines of silicon provided 90 measurements for least-squares computations.

Standards used:

$$\lambda(\text{Cu } K\alpha_1) = 1.54051 \text{ \AA}$$

$$\lambda(\text{Cu } K\alpha_2) = 1.54433 \text{ \AA}$$

For Si line 533:

$$\theta(\text{Cu } K\alpha_1) = 68.446^\circ, \theta(\text{Cu } K\alpha_2) = 68.811^\circ$$

\* Standardized Si powder was obtained from Dr W. Parrish, Philips Laboratories, Irvington-on-Hudson, New York, U.S.A.

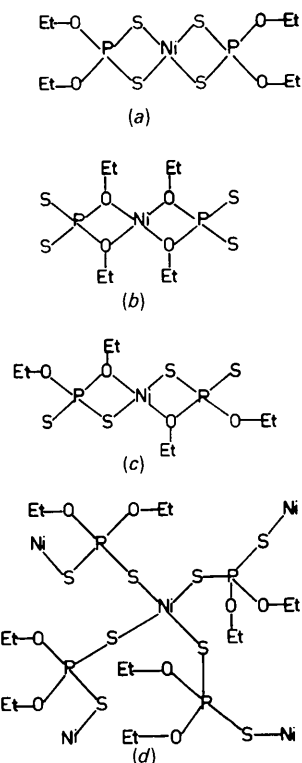


Fig. 1. Possible structural configurations for  $\text{Ni}[(\text{C}_2\text{H}_5\text{O})_2\text{PS}_2]_2$ . (a) is considered most likely on chemical grounds.

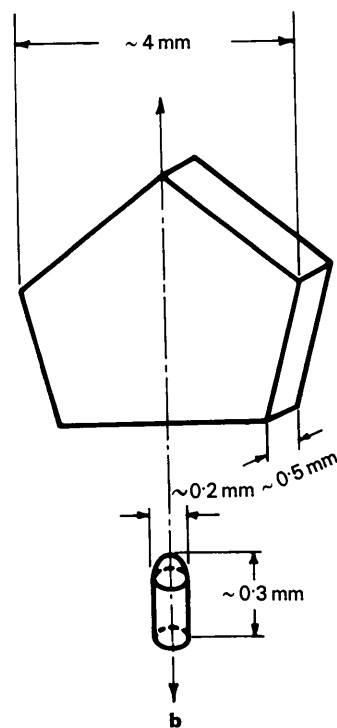


Fig. 2. The shape and size of the initial and final crystals.

For Si line 444:

$$\theta(\text{Cu } K\alpha_1) = 79.314^\circ, \theta(\text{Cu } K\alpha_2) = 80.100^\circ$$

Crystals used for the collection of intensity data were obtained by slow evaporation of an acetone solution at a few degrees below  $0^\circ\text{C}$  and had a plate-like habit with one dimension an order of magnitude smaller than the other two. Such a habit and the considerable value of  $\mu$  tended to produce doubling of some spots. This effect was minimized by dissolving a large crystal down to a roughly cylindrical shape. The shapes and sizes of the initial and final crystals are shown in Fig. 2. The spots still showed a trace of doubling and it was decided to integrate the intensities in one direction (rotation of the film cassette) in order to facilitate visual measurements. (This procedure does not, of course, correct the absorption error.)

Using a Nonius integrating Weissenberg camera and the multiple film technique, 8 layers of intensity photographs,  $h0l$  through  $h7l$ , were measured visually with the aid of a calibration strip and corrected for Lorentz and polarization effects. Another similar crystal was used to obtain  $hk0$  intensities to provide approximate inter-layer scale factors for a Patterson synthesis.

All photographs were taken at room temperature ( $20^\circ\text{C}$ ). A total 1411 reflexions were measured, including 214 unobserved.

No corrections were made for absorption or extinction, but a few strong reflexions were omitted from the least-squares refinement. During an early stage of investigation, rotation and zero-layer Weissenberg photographs about the same crystallographic direction were prepared both for  $\text{Ni}[(\text{OEt})_2\text{PS}_2]_2$  and also for  $\text{Pd}[(\text{OEt})_2\text{PS}_2]_2$ . The diffraction patterns coincided on superposition, confirming chemical opinion that these two compounds are isomorphous. No attempt was made to pursue this aspect any further.

### Structure determination

From a three-dimensional unsharpened Patterson synthesis, peaks consistent with a square-planar Ni-S-P-S arrangement were found. Structure factors calculated up to  $\sin^2\theta = 0.3$ , using the coordinates of these four atoms, gave a reliability index  $R = 0.46$ . A three-dimensional Fourier synthesis based upon these structure factors appeared encouraging and gave more satisfactory coordinates for the Ni, S and P atoms. Initially an isotropic temperature factor  $B = 2.0 \text{ \AA}^2$  was assumed, which was later adjusted to  $3.5 \text{ \AA}^2$ . The oxygen atoms were located by correlation of the peaks in successive Fourier syntheses with Patterson peaks; the carbon atoms were located with less certainty. The reliability index was 0.34 at this stage. The positions of the carbon atoms were not located with any certainty until the refinement stage, when the full range of intensity data was being used, and the heavier atoms were near to their final positions. After several three-dimensional difference syntheses and one cycle of a differential synthesis,  $R$  was reduced to 0.25 for all reflexions excluding unobserved.

At this stage refinement by least-squares was commenced. Unobserved reflexions where  $F_c$  was less than the minimum observable were excluded from the refinement, as were other reflexions affected by extinction or whose observed values were considered unreliable. These are indicated by an asterisk in Table I. Computations were carried out on UTECOM, a DEUCE computer, using a least-squares program written by Rollett (1961), modified by B. McHugh and R. Vowels of the University of New South Wales Computing Department.

Eleven cycles of refinement reduced the  $R$  index to 0.124 excluding unobserved reflexions and those marked with an asterisk in Table I, which lists observed

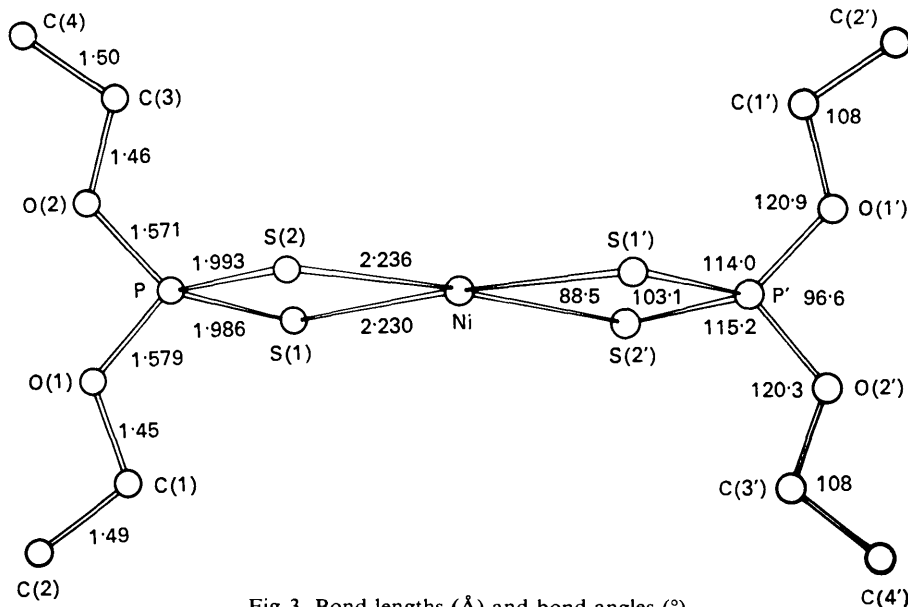


Fig. 3. Bond lengths ( $\text{\AA}$ ) and bond angles ( $^\circ$ ).



and calculated structure factors ( $25 \times$  absolute scale). During this stage of refinement the results were re-scaled, layer by layer, after each cycle. The final calculations were carried out on the IBM 7040 computer

at the Australian Atomic Energy Research Establishment at Lucas Heights. Two more cycles of a full-matrix least-squares refinement, using a program prepared by Mrs Hogg (Hogg, 1965) reduced the residual

Table 2. Positional and thermal parameters for the atoms in the asymmetric unit

The temperature factors are in the form:  
 $T_i = \exp [-(B_{11}h^2 + B_{22}k^2 + B_{33}l^2 + 2B_{12}hk + 2B_{13}hl + 2B_{23}kl)]$   
 All parameters have been multiplied by  $10^4$

	$x(\sigma_x)$	$y(\sigma_y)$	$z(\sigma_z)$	$B_{11}(\sigma_{B_{11}})$	$B_{22}(\sigma_{B_{22}})$	$B_{33}(\sigma_{B_{33}})$	$B_{12}(\sigma_{B_{12}})$	$B_{13}(\sigma_{B_{13}})$	$B_{23}(\sigma_{B_{23}})$
Ni	0	0	0	71 (2)	56 (4)	174 (5)	3 (2)	34 (3)	-9 (3)
S(1)	1876 (3)	1024 (4)	21 (5)	61 (3)	23 (6)	211 (8)	-11 (3)	52 (4)	-28 (5)
S(2)	771 (4)	-1707 (4)	-1093 (6)	81 (4)	21 (6)	221 (8)	1 (3)	66 (4)	-25 (5)
P	2337 (3)	-586 (4)	-983 (4)	49 (3)	34 (5)	128 (6)	16 (3)	30 (3)	5 (4)
O(1)	2689 (8)	-357 (10)	-2625 (11)	70 (9)	76 (14)	111 (14)	28 (8)	25 (9)	31 (11)
O(2)	3673 (9)	-1221 (11)	-166 (11)	71 (9)	48 (14)	140 (16)	34 (9)	34 (10)	25 (11)
C(1)	1758 (17)	255 (19)	-3882 (22)	116 (18)	68 (27)	187 (32)	44 (17)	0 (19)	33 (23)
C(2)	2333 (20)	291 (20)	-5293 (21)	186 (25)	95 (27)	157 (27)	0 (20)	33 (21)	24 (21)
C(3)	3972 (17)	1497 (19)	1507 (18)	130 (20)	110 (27)	133 (24)	65 (17)	54 (18)	33 (19)
C(4)	5095 (15)	-2426 (18)	1840 (21)	72 (15)	109 (27)	198 (30)	29 (15)	9 (17)	81 (21)

Table 3. Interatomic distances and bond angles, and their standard errors

Ni—S(1)	$2.230 \pm 0.004$ Å	S(1)—Ni—S(2)	$88.5 \pm 0.1^\circ$
Ni—S(2)	$2.236 \pm 0.004$	S(1)—P—S(2)	$103.1 \pm 0.2$
S(1)—P	$1.986 \pm 0.006$	S(1)—P—O(1)	$114.0 \pm 0.4$
S(2)—P	$1.993 \pm 0.005$	S(2)—P—O(2)	$115.2 \pm 0.5$
P—O(1)	$1.579 \pm 0.009$	O(1)—P—O(2)	$96.6 \pm 0.5$
P—O(2)	$1.571 \pm 0.009$	P—O(1)—C(1)	$120.9 \pm 0.9$
O(1)—C(1)	$1.45 \pm 0.02$	O(1)—C(1)—C(2)	$108 \pm 1$
C(1)—C(2)	$1.49 \pm 0.02$	P—O(2)—C(3)	$120.3 \pm 0.8$
O(2)—C(3)	$1.46 \pm 0.02$	O(2)—C(3)—C(4)	$108 \pm 1$
C(3)—C(4)	$1.50 \pm 0.02$		

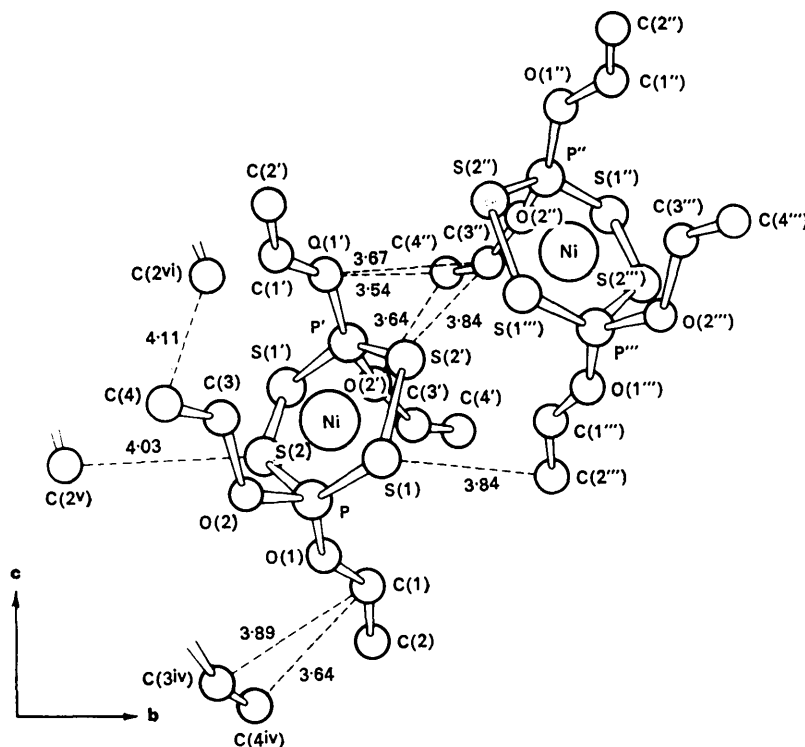


Fig. 4. Intermolecular distances and the relative positions of the two molecules in the unit cell. (Lines joining sulphur atoms do not represent bonds.)

to 0.115. The function minimized in the least-squares refinement was  $\sum w(|F_o(hkl)| - |F_c(hkl)|)^2$ . Two weighting schemes were used. For the earlier cycles

$$\begin{aligned} \sqrt{w} &= |F_o|/|F^*| \text{ for } |F_o| \leq |F^*| \\ \sqrt{w} &= |F^*|/|F_o| \text{ for } |F_o| > |F^*| \end{aligned}$$

where  $|F^*|$  is a selected value. In this case, the value of  $|F^*|$  chosen was 60 electrons, which is approximately one-half of the maximum structure factor observed, and so this scheme gives highest weight to reflexions of moderate intensity. For the last cycles, the weighting was

$$\begin{aligned} \sqrt{w} &= 1 \quad \text{for } |F_o| \leq |F^*| \\ \sqrt{w} &= |F^*|/|F_o| \text{ for } |F_o| > |F^*|. \end{aligned}$$

which increased the weight for weaker reflexions.

Four hydrogen atoms attached to C(1) and C(3) (for numbering see Fig. 3) were placed in the expected tetrahedral positions, and reduced  $R$  by 0.004, but when they were included in a least-squares cycle their indicated shifts were meaningless. It was concluded that the data were not sufficiently good to locate hydrogen atoms and they were removed from the final calculations. Atomic scattering factors were taken from *International Tables for X-ray Crystallography* (1962) for all atoms. Anisotropic temperature factors were used for the last 8 cycles of refinement on UTECOM and the 2 cycles on the IBM 7040 computer. Positional and thermal parameters together with their standard deviations are listed in Table 2. The  $B_{ij}$ 's are coefficients in the expression

$$\exp[-(B_{11}h^2 + B_{22}k^2 + B_{33}l^2 + 2B_{12}hk + 2B_{13}hl + 2B_{23}kl)]$$

Interatomic distances, bond angles and thermal ellipsoids were calculated on the IBM 7040 computer using the ORFFE program by W.R. Busing, K.O. Martin and M.A. Levy of Oak Ridge National Laboratory, U.S.A. The interatomic distances and bond angles are shown in Fig. 3 and are listed in Table 3.

### Discussion

A difference synthesis calculated with the use of the final parameters showed a negative peak of  $3.00 \text{ e.}\text{\AA}^{-3}$  at the site of the nickel atom, perhaps owing to the neglect of the rather large anomalous dispersion correction ( $\Delta f' = -3.2$  electrons; Cromer, 1965). Elsewhere the maximum variation was  $\pm 0.6 \text{ e.}\text{\AA}^{-3}$ . At the sites of all atoms (excluding nickel) the value of  $\Delta \rho$  was approximately zero.

Bond lengths agree very well with similar lengths reported in other structures. For example the Ni-S distances obtained in this work (2.23 and 2.24 Å) are in exact agreement with those in nickel(II) xanthate (Franzini, 1963) and are in the range 2.1–2.3 Å found for other square-planar nickel(II) complexes. In the organic part of the molecule, the mean O-C and C-C distances of 1.45 and 1.49 Å are normal for this type of compound. For example, in  $[\text{C}_2\text{H}_5\text{O}_4\text{S}]^-$  reported

by Jarvis (1953) the corresponding distances were 1.44 and 1.51 Å.

The intermolecular distances as well as the relative positions of the two molecules in the unit cell are shown in Fig. 4. These are consistent with packing by van der Waals forces, and show no tendency for the nickel atom to coordinate to any atoms in another molecule.

The anisotropic temperature coefficients,  $B_{ij}$  show no unexpected features. However, Lingafelter & Donohue (1966) have pointed out that interaction occurs between temperature and scale factors when data about one axis only are being used, so that caution is required when interpreting the thermal parameters, particularly  $B_{22}$ . However, the favorable correlation matrix, and the fact that the scale factor was not freely variable, have convinced the authors that the thermal ellipsoids shown in Fig. 5 are of semi-quantitative significance. These indicate that the largest r.m.s. vibrational components are in the directions of smallest constraint and *vice versa*. For example, Ni, S and P all have a large component normal to their plane; the C and O atoms tend to vibrate least along the bond direction.

The phosphorus atom is practically in the Ni-S plane, its calculated distance from the plane being 0.03 Å. The plane through O(1), O(2) and P is almost at right angles to the Ni-S plane, the calculated angle being  $87.9^\circ$ . This plane also passes through the Ni atom, within the limits of observational error. The bond distribution around P is therefore approximately tetrahedral. Atoms C(2) and C(4) also lie in the O-P-O plane within the limits of experimental error, the calculated distances from the plane being 0.02 Å and 0.04 Å respectively.

The authors are indebted to Professor S.E. Livingstone, Chemistry School, University of New South

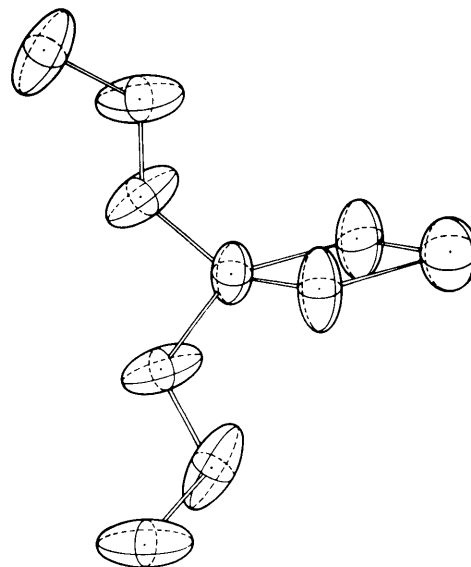


Fig. 5. Vibrational ellipsoids.

Wales, for suggesting this problem and for many helpful discussions.

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## A Refinement of the Parameters of $\alpha$ Manganese

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The atomic parameters of  $\alpha$  manganese, first determined by Bradley & Thewlis (1927), have been refined from an  $R$  index of 0.093 to a final  $R$  of 0.029 for fifty reflections. This refinement was based on an analysis of the integrated intensities of X-ray diffracted powder peaks measured independently at two laboratories. The lattice constant of  $\alpha$  manganese was determined from liquid helium temperature up to room temperature and a change in the thermal expansion coefficient was found in the vicinity of the Néel temperature (95°K) although no structural change was observed in this region.

The structure of  $\alpha$  manganese as determined by Bradley & Thewlis (1927) is as follows:

Space group  $I\bar{4}3m$  no. 217 in *International Tables for X-ray Crystallography* (1952).

2 Mn(1) in 2(a) 000, etc.  
 8 Mn(2) in 8(c) xxx, etc.  $x(2)=0.317$   
 24 Mn(3) in 24(g) xxz, etc.  $x(3)=0.356$ ,  $z(3)=0.042$   
 24 Mn(4) in 24(g) xxz, etc.  $x(4)=0.089$ ,  $z(4)=0.278$

This result is based on X-ray intensities read from Phragmén-camera photographs taken by Westgren & Phragmén (1925). Kasper & Roberts (1956) checked Bradley & Thewlis's parameters with neutron diffraction data and found them to be accurate enough ( $\pm 0.003$ ) to establish the magnetic structure at 4.2, 77, and 298°K.

In an attempt by Gazzara, Middleton & Weiss to determine the characteristic temperature  $\theta$  of  $\alpha$ -Mn it was found desirable to refine the structural parameters by applying modern experimental X-ray diffraction techniques. A refinement was performed on twenty-one Cr  $K\alpha$  diffraction peaks from powder specimens. During this same time, a similar refinement was performed by E. O. Hall on  $\alpha$ -Mn powders using Cu  $K\alpha$

radiation. This report presents the best values of the structural parameters obtained from the two investigations.

Although many powder samples were examined in this X-ray diffraction analysis, the data from only four sources were eventually used:

(1) A pure, carbon-free, electrolytic -400 mesh powder. This showed some diffraction peak broadening and gave intensities identical with those of sample (2).

(2) Electrolytic plate, ground mortar and pestle and sifted to -400 mesh particle size. Although the cold working increased the diffraction peak widths, it reduced extinction effects. The integrated intensity of the Cr  $K\alpha$  diffraction peak  $\Sigma h^2=18$ , for example, was increased 4% through cold working. All calculations involving Cr  $K\alpha$  radiation are based on measurements performed on this material.

(3) A portion of powder sample (2) annealed at 440°C for 90 hours in vacuum. All of the diffraction peaks obtainable with Cr  $K\alpha$  radiation from this sample could be resolved and accurately measured with Mo  $K\alpha$  radiation; many of the high-angle Mo  $K\alpha$  diffraction peaks were resolved sufficiently well to be integrated. All results obtained with Mo  $K\alpha$  radiation were taken